\$/133/62/000/009/005/009 A054/A127

AUTHORS:

Kolmogorov, V.L., Candidate of Technical Sciences, Selishchev,

K.P., Engineer

TITLE:

Cold drawing of tubes under improved lubricating conditions

PERIODICAL:

Stal', no. 9. 1962, 830 - 831

TEXT: Tests were carried out to improve the lubrication in drawing tubes without using mandrels. For this purpose a simple device consisting of a sleeve and a finely dispersed clean high-viscosity sodium soap powder were used. Tubes of "20" and 1 X 18 H 9 T (IKh 18 N) T) grade steel were drawn with a wall-thickness-to--diameter ratio varying between 0.05 and 0.13, at drawing rates of 0.17 - 0.58 m/ /sec. The soap powder applied forms a dense, glassy, adhesive coating, 0.007 -0.031 mm thick, on the tube surface, which is sufficient to prevent any direct contact between the drawing die and the tube surface. When this new lubrication method is applied, the service life of the drawing tool will be raised considerably; moreover, stainless steel tubes can be drawn by dies of 12 X5MA (12Kh5MA) steel instead of "pobedit" (sintered carbon); the drawing power required will be reduced by 27 - 29%, and stainless steel tubes of a higher surface quality can be

Card 1/2

Cold drawing of

3/133/62/000/009/005/009 A054/A127

produced. An essential condition of using sodium soap powder as lubricant is that the tube surface must be dried carefully prior to drawing. There is I figure.

ASSOCIATION:

Ural'skiy nauchno-issledovateľskiy institut chernykh metallov (Ural Scientific Research Institute of Ferrous Metals)

Card 2/2

KOLMOGOROV, V.L.; ORLOV, S.I.; SELISHCHEV, K.P.; LEKARENKO, Ye.M. [deceased]; POKROVSKAYA, G.N.; TIKHONOV, D.Ya.; BOGOMOLOV, I.F.

Drawing wire of nonferrous metals and alloys in conditions of fluid friction. TSvet. met. 36 no.12:65-67 U 163. (MIRA 17:2)

"APPROVED FOR RELEASE: 08/23/2000 C

CIA-RDP86-00513R001547720011-3

L 59k81-65 EFF(c)/EMT(m)/EMP(k)/EMA(c)/EMF(b)/T/EMF(v)/EMP(t) Pr-L/Pr-L DJ/

JM/JD/HM
ACCESSION NR: AR5015177 UR/0137/65/000/005/D035/D035

SOURCE: Ref. zh. Metallurgiya, Abs. 5D208

AUTHOR: Kolmogorov, V. L.; Selishchev, K. P.; Orlov, S. I.

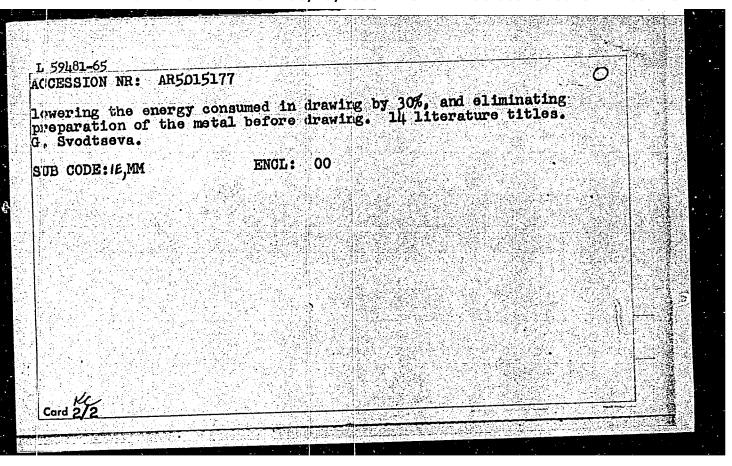
TITIE: Drawing of tubes, rods, and wire under conditions of hydrodynamic friction

CITED SOURCE: Tr. Ural'skogo n.-1. in-ta chern. met., v. 3, 1964, 64-81

TOPIC TAGS: drawing, tube, rod, wire, hydrodynamics, friction, hydrodynamic friction, rheological property, lubricant

THANSLATION: The article presents the results of an industrial test of drawing tubes and wires under conditions of hydrodynamic friction with the use of dies, the results of a study of the rheological properties of dies, the results of a study of the rheological properties of lubricants, and the development of a theory of drawing under conditions of hydrodynamic friction. Drawing under conditions of hydrodynamic friction permits increasing the speed of drawing, increasing the durability of the instrument by 11 times,

Card 1/2



3-7-10/29 Selishchev, V.I. AUTHOR &

On Some Possibilities of Improving Technical Training in Correspondence and Evening Courses (O nekotorykh vozmozhnost-TITLE

yakh uluchsheniya zaochnogo i vechernego tekhnicheskogo

obrazovaniya)

ABSTRACT:

Vestnik Vysshey Shkoly, 1957, # 7, pp 45-49 (USSR) PERIODICAL:

The author states that a characteristic fact in the devel-

opment of Soviet higher schools is the considerable increase in student training without interrupting the students' industrial work. As an example the author quotes the system of correspondence schooling used by the Ministry of Public Transport which includes the All-Union Correspondence Institute for Railway Transport Engineers (VZIIT), 4 correspondence sections in day vuzes and 26 consultation centers all serving about 20,000 students distributed among 120 railroad centers. The permanent teaching staff, which includes 17 professors and 69 dotsents, is concentrated primarily in the chairs of the VZIIT in Moscow, while about 1,000 part-time teachers - paid by the hour - are used. The author then mentions the shortage

of laboratories and qualified teachers in the correspondence school system. He then suggests creating general engineering Card 1/3

3-7-10/29

On Some Possibilities of Improving Technical Training in Correspondence and Evening Courses.

and general economics faculties in the existing vuzes for the first years of correspondence training. At the end of the third year, the assignment of the students to certain institutions should be made in accordance with their special ties.

The assignment to a specialized school after the third year should not mean a transfer to a remote institute of the special branch, since special faculties may be reorganized in any nearby related technical vuz. He mentions several vuzes where this has been done and several where it could be done and adds that it must be understood that the suggested reorganization should be applied also to methodical management. In this connection the vuz bases will play an important part. For many special ties one important vuz base would be sufficient; however, in some cases, methodical management could be assigned to vuzes situated in different economic areas.

Finally the author mentions some of his observations relating to evening courses in technical vuzes which are appropriate for workers in industrial enterprises but are not appropriate for students employed, for example, as travelling railroad

Card 2/3

3-7-10/29

On Some Possibilities of Improving Technical Training in Correspondence and Evening Courses.

le ant-ratification alla de deservirell, som enkism**e** discrete et l

workers.

Moreover, students who do not comply with the requirements of day courses, often take refuge in evening courses. As an example the author mentions the evening courses of the Moscow Transport Economics Institute where only 59 out of 720 students are actual transport workers while 398 do not work at all. These students reach the VIth year's course without any practical experience. As a result the institute often releases unqualified specialists.

AVAILABLE

Library of Congress

Card 5/3

AUTHOR:

Selishchev, V.I.

SOV/3-58-11-11/38

TITLE:

The Form of Industrial Training is Changing (Menyayetsya

soderzhaniye proizvodstvennogo obucheniya)

PERIODICAL:

Vestnik vysshey shkoly, 1958, Nr 11, pp 30 - 34 (USSR)

ABSTRACT:

In the near future, the number of higher school students will be reinforced by youth having professional and engineering skill. It will, therefore, not be necessary to begin industrial training with the rudiments. This does not mean that measures for improvement of industrial training should be postponed until the reorganization of the secondary school is completed. These two processes should take place simultaneously. The author sets forth his suggestions for the improvement of the industrial training of those students who have no experience. He begins with the onthe-job-training workshops established at the vuzes where the organization of work is poor. As an example of good organization of practical training, the author mentions the Moskovskiy institut inzhenerov zheleznodorozhnogo transporta (Moscow Institute of RR Engineers) and those in Rostov and Dnepropetrovsk, where the internal routine work has been brought close to production conditions. He speaks of the advisability of replacing the pre-diploma practice in the

Card 1/2

The Form of Industrial Training is Changing

SOV/3-58-11-11/38

curricula by an obligatory 6 months' industrial practice before submitting the graduation thesis. A change in the form . of industrial practice is already taking place in such transport vuzes as the Leningrad Institute of RR Engineers, where 72 % of the entire number of probationers are working independently, and this is also the case at the Moscow, Tomsk, Novosibirsk and Tashkent institutes. Practical work was especially well organized at the locomotive and car manufacturing plants. He also speaks of the difficulty of finding plants where practical training could take place.

ASSOCIATION: GUUZ Ministerstva putey soobshcheniya SSSR (GUUZ, USSR Ministry of Transportation)

Card 2/2

Track machinery station for student training. Put' i put. khoz.

(MIRA 15:10)

(Railroads—Track)

(Railroad engineering—Study and teaching)

L 23571-66 EWT(d)/EWP(c)/T/EWP(v)/EWP(h)/EWP(h)/EWP(l)

ACC NR: AP6002600 (A) SOURCE CODE: UR/0286/65/000/023/0095/0095

AUTHORS: Selishchev, Ye. M.; Pashteyn-Sitnikov, N. V.; Volkernyuk, V. V.

ORG: none

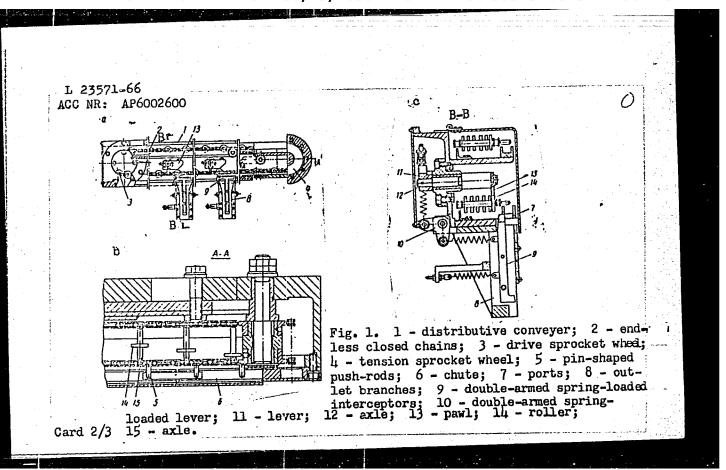
TITLE: Distributive conveyer for <u>automated lines</u>. Class 81, No. 176825 / announced by Special Construction and Technological Bureau for Design of Metal-Cutting Tools and Equipment (Spetsial noye konstruktorskoye i tekhnologicheskoye byuro proyektirovaniya metallorezhushchego instrumenta i oborudovaniya)

SOURCE: Byulleten' izobreteniy i tovarnykh znakov, no. 23, 1965, 95

TOPIC TAGS: conveying equipment, automation equipment

ABSTRACT: This Author Certificate presents a distributive conveyer for automated lines. Endless closed chains are mounted in the frame of the conveyer and are engaged with drive and tension sprocket wheels. To simplify the design and to increase the operation reliability with various technological handling processes, one of the chains carries pin-shaped push-rods on its outer edge (see Fig. 1). A chute with distributive ports for outlet branches is mounted under the push-rods in the frame of the conveyer. The ports are closed by double-armed spring-loaded

Card 1/3 UDC: 621.867



L 23571-66 ACC NR: AP6002600

interceptors which are linked through a system of spring-loaded levers to pawl axles fastened to the frame. During operation of the conveyer the pawls interact with rollers placed on axles mounted between the chains in front of the corresponding push-rods. Orig. art. has: 1 diagram.

SUB CODE: 13/

SUBM DATE: 06Apr64

Card 3/3

ATAMASIU, Al., ing.; SELISCHI, N., ing.; LUPSE, T., ing.

Present problems relating to road maintenance. Rev transport

9 no.5:219-221 My '62.

SELISKAR, Ruze

Mineral resources of Algeria, Geogr obz 8 no. 3/4:100 '61.

SELISKAR, S

Development of machine printing in Yugoslavia. p. 622. TEKSTIL. Vol. 4. No. 6, June 1955. Beograd.

SCURCE: East European Accessions List (EEAL), Library of Congress, Vol. 4, No. 12, December 1955.

SELISKO G Country :GDR :Organic Chemistry. Synthetic Organic Chemistry Casecory No. 15317 : Ref Zhur - Khim., No 5, 1959, Abs. Jour : Selisko, O.; Schubert, A. Author Institut. :On Substances with Anti-E-Action. Report III. Title Certain New Compound Ethers of Phenols :Ernaehrungsforschung, 1958, 3, No 2, 224-226 Orig Pub. :In continuation of studies begun earlier (re-Abstract port II, see Ref Zhur-Khim, 1958, 43281), a series of compound ethers (CE) of phenols was synthesized. 35 g. of o-(I), m-(II) or p-cresol, 12.5 g. of NaOH and 150 ml. of ChHoOH are boiled for 30 minutes, the water is distilled off, 44 g. of ClCH2COOC2H7 are slowly added, boiled for 15 minutes, 14 g. of NaOH in 150 ml. of water are added as rapidly as possible, boiled for another 15 minutes, evaporated, and 1/5 Card:

G Sountry Catogony No. 15317 : Ref Zhur - Knim., No 5, 1959, , bs. Jour Author Institut. Title Orig Pub. :2-, 3-, or $4-\text{CH}_3\text{C}_6\text{H}_4\text{OCH}_2\text{COOH}$ (III) is precipitated by HCl, with a yield of more than 70%.

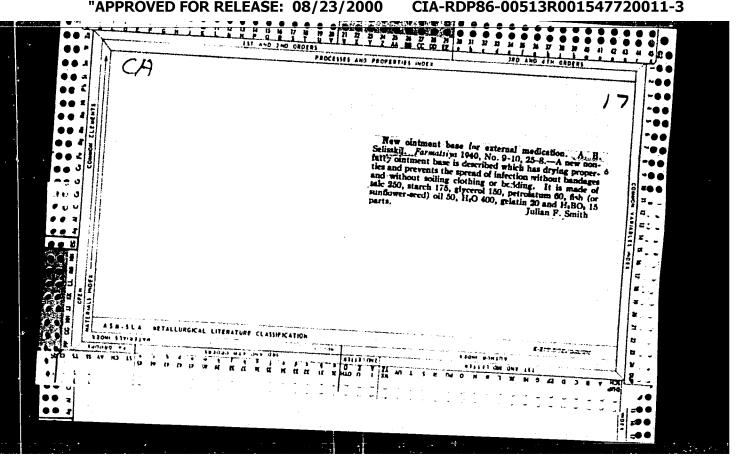
17 g. of III, 11 g. of II and 8 g. of POCl₃ are heated at 110-140 up to cessation of the separation of HCl, cooled to 40°, added to the ex-Abstract cont'd. cess of the solution of NaHCO3, and x-CH3C6H14O- $CH_2COOC_6H_LCH_3-y$ (IV) (x = 3, y=2) (IVa), b.p. 175-1760/1 mm. are extracted with ether. Analogously, other IV are obtained (x and y, tem-2/5 durá: G - 10

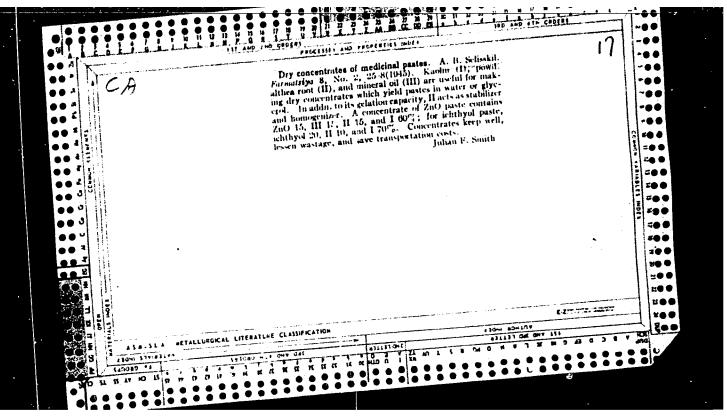
		G	
	Country Category Abs. Jour Author Institut.	: Ref Zhur - Khim., No 5, 1959, No. 15317	
•	Oriz Tub. Abstract	:perature of reaction, m.p. in °C. are given): 2, 4, 150-155, 56-57; 3, 2, 90-120, - (b.p. 178-180°/1 mm.); 3, 3, 130-140, - (b.p. 194- 196°/1 mm.); 3, 4, 110-120, 59-60; 4, 2, 140- 196°/1 mm.); 3, 4, 110-120, 59-60; 4, 2, 140- 150, 37.5-38.5 (b.p. 168-170°/1 mm.); 4, 3, 150, 37.5-38.5 (b.p. 168-170°/1 mm.); 4, 3, 150, 37.5-38.5; 4, 4, 130-140, 126-126.5. 135-140, 71-71.5; 4, 4, 130-140, 126-126.5. 40 g. of CH ₃ CH=CHCOCl (V, VI acid) and 35 g. 40 g. of CH ₃ CH=CHCOCl (V, VI acid) and and poured into a solution of NaHCO ₃ , and and poured into a solution of NaHCO ₃ , and CH ₃ CH=CHCOOC ₆ H ₄ CH ₃ -2 is extracted with ether,	

c:	Suntry : atogory : Ref Zhur - Khim., No 5, 1959, No. 15317	
T	uthor : nativut. : itlo :	
	bstract contid. b.p. 125-127°/12 mm. Analogously, from 22 v, CE of VI are obtained (original phenol quantity in g., reaction temperature in contide the contide of the con	are arva- OCH3-2, 40 g. with
	Carã: 4/5	
	c = 11	

SELISSKAYA, Ye.A.; OSTRAYA, S.S.

Blackheads and pimples in infants. Vest. derm. i ven. 33 no.2: 82-83 Mr-Ap 159. (MIRA 12:7)





COLICSKIY, Frof. A. B.

Nor., Clinic Skin Diseases, Central Syphilodernatological Inst., Min. Health, -1947...
"Fenicillin Salve for the Treatment of Some Skin Diseases," Vest. Venerol. i Dernatol.,
No. 5, 1947; "Penicillin Ointment for Treating Skin Diseases," Fol'deher i Akusher.,
No. 1, 1948.

SELISSKIY. A. B. Prof

FA 41T65

USSR/Medicine - Skin Diseases

Jan 1948

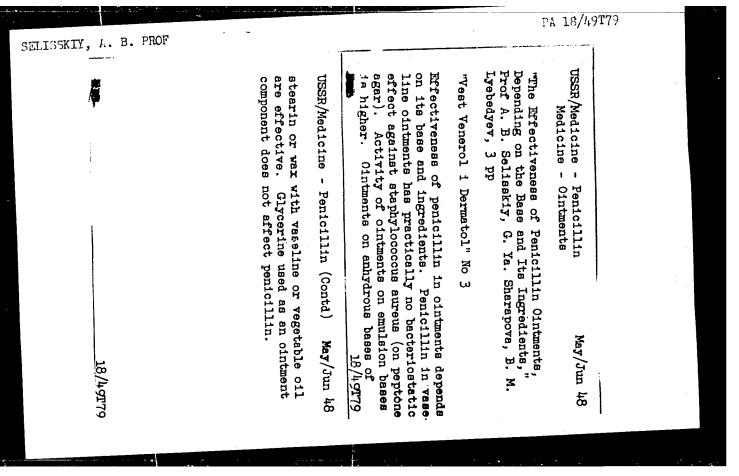
Medicine - Penicillin

"Penicillin Ointment for Treating Skin Diseases," Prof A. B. Selisskiy, 3 pp

"Fel'dsher i Akusherka" No 1

Penicillin ointment is very effective in coping with pyococcus infection of the skin. In some instances (furuncles, carbuncles, etc.) treatment required a combined form, i.e., penicillin ointments on the skin surface and penicillin injections.

41765



SELECKLY, A.F.

LEHEDEV, B. N., SELISSKII, A. B.

Selebin, a new preparation for treatment of eczema. Vest. vener. No. 4, July-Aug. 50. p. 41

1. Of the Skin Clinic (Head-Prof. L. N. Mashkilleyson), Central Skin-Venereological Institute (Director-Candidate Medical Sciences N. M. Turanov) of the Ministry of Public Health USSR.

CLML 19, 5, Nov., 1950

(MLRA 8:11)

SELISSKIY, A.B. [Treating skin diseases and prescribing for them; manual for physicians] Lechenie zabolevanii kozhi i retseptura; spravochnik dlia vrachei. Minsk, Izd.-vo Akademii nauk BSSR, 1955.

271 p.

(SKIN---DISEASES)

SELISSKIY, A.B.

Cutaneous nerves in eczema. Arkh.pat. 17 no.1:68 Ja-Mr '55
(ECZEMA, physiology,
skin nerves)
(SKIN, innervation,
in eczema)

SELISSKIY, A.B.

[Skin diseases in children and adolescents] Bolezni kozhi u detei
podrostkov. Minsk, Izd-vo Akademii nauk BSSR, 1957, 269 p.
(SKIN-DISEASES)

(MIRA 10:7)

SELISSKIY, Aleksandr Borisovich

[Manual on skin diseases; clinical aspects and treatment of skin diseases, prescription writing, homeotherapy] Spravochnik po kozhnym bolezniam; klinika i lechenie zabolevanii kozhi, retseptukozhnym bolezniam; klinika i lechenie zabolevanii klinika i leche

Dermabrasion; survey of the literature. Vest.derm.1 ven. 34 (MIRA 13:11) no.10:35-39 160. (SKIN-SURGERY)

SELISSKIY, Aleksandr Borisovich, prof.; PAVLOV, N.F., dots., red.; ZAYTSEVA, T., red. izd-va; VOLOKHANOVICH, I., tekhn. red.

[A guide to skin diseases; clinical aspects and treatment of diseases of the skin, pharmacotherapy and prescription filling] Spravochnik po kozhnym bolezniam; klinika i lechenie zabolevanii kozhi, farmakoterapiia i retseptura. Izd.2., pereridop. Minsk, Izd-vo Akad.nauk BSSR, 1961. 412 p. (MIRA 15:1)

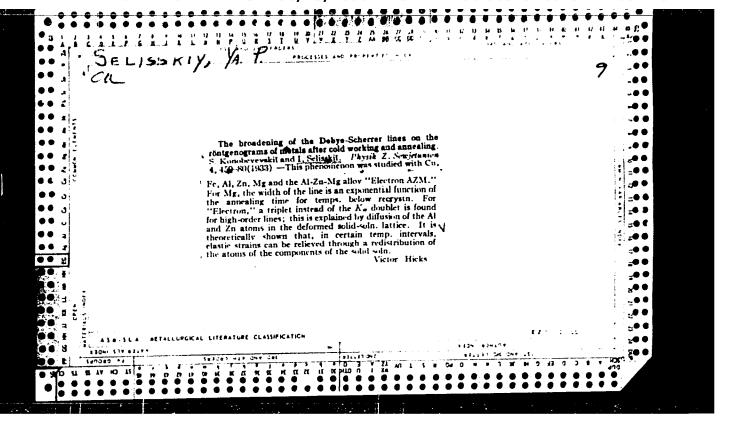
(SKIN-DISEASES)

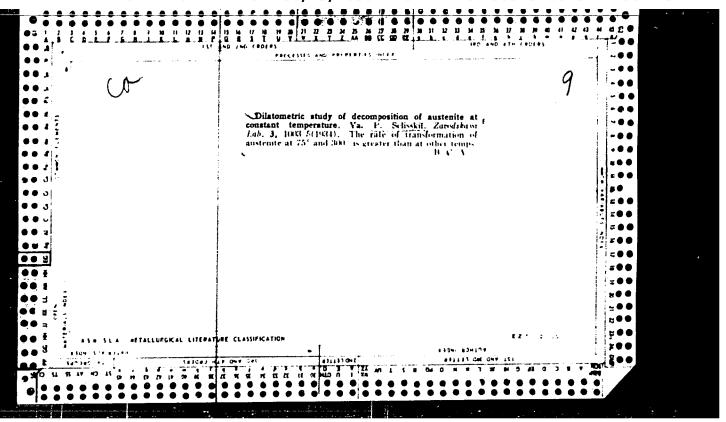
SELISSKIY, Aleksandr Borisovich, prof.; POTEYENKO, M., red.; VARENIKOVA, V., tekhn. red.

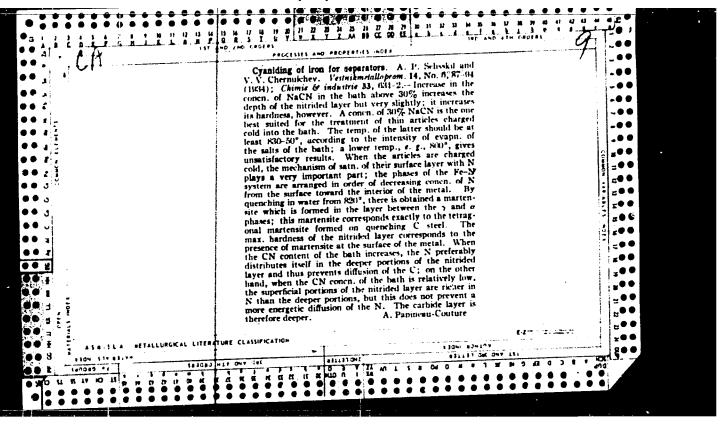
0

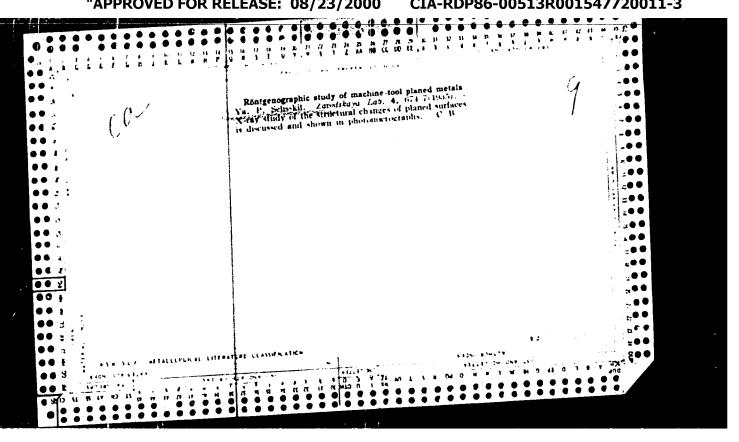
[Marual on skin diseases; clinical aspects and treatment of skin diseases, pharmacotherapy and prescription] Spravochnik po kozhnym bolezniam; klinika i lechenie zabolevanii kozhi, farmakoterapiia i retseptura. Izd.3., perer. i dop. Minsk, Gosizdat BSSR, 1963. 475 p. (MIRA 17:2)

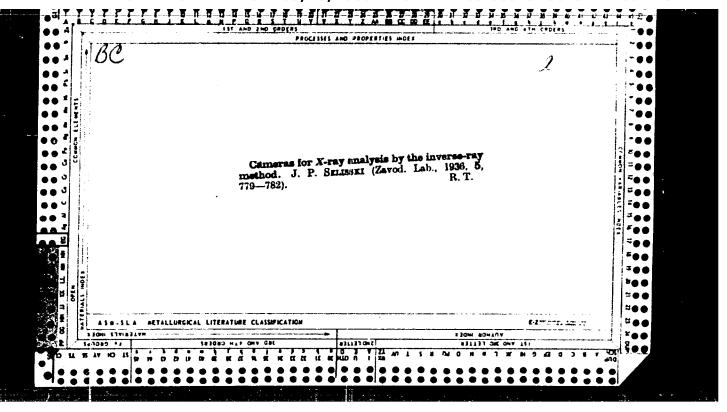


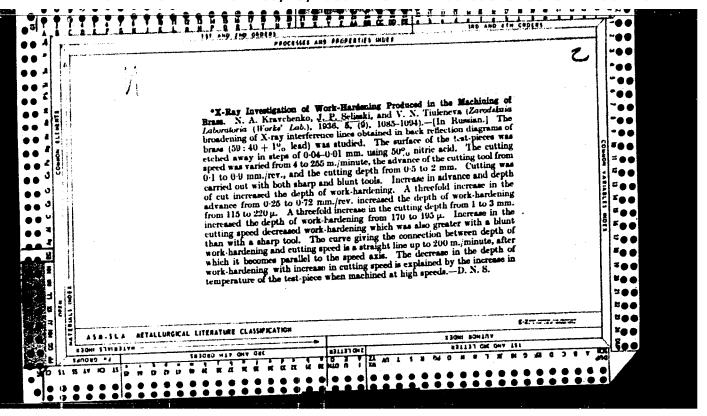


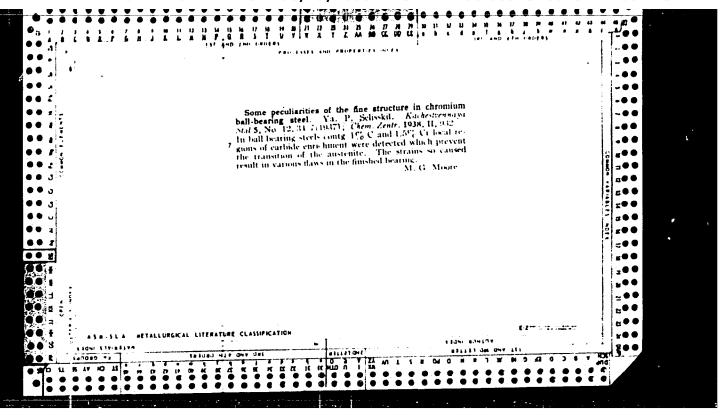


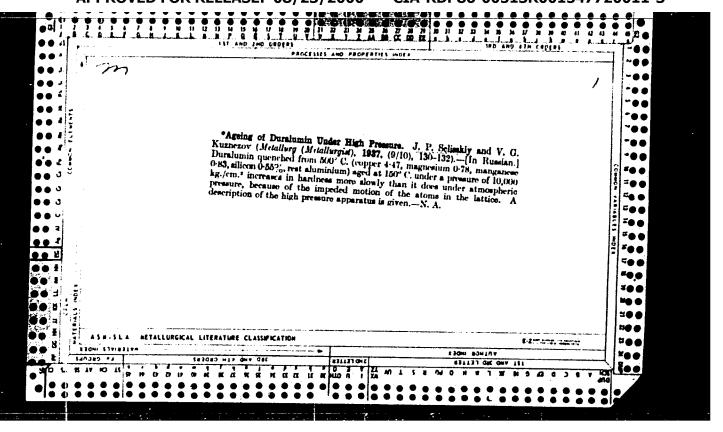


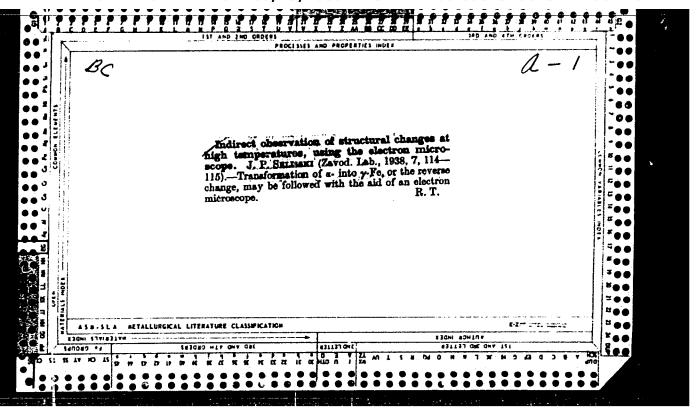


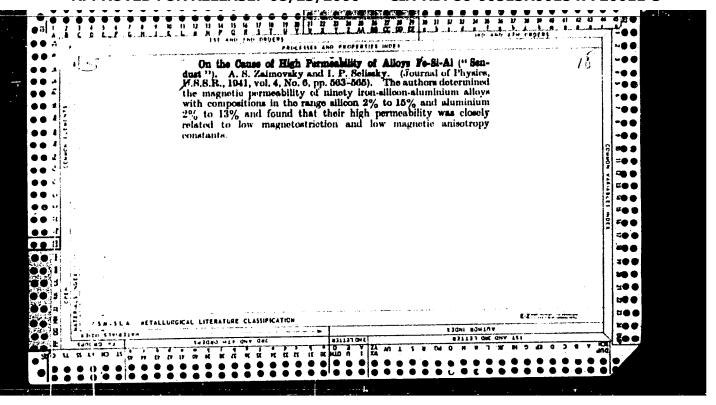


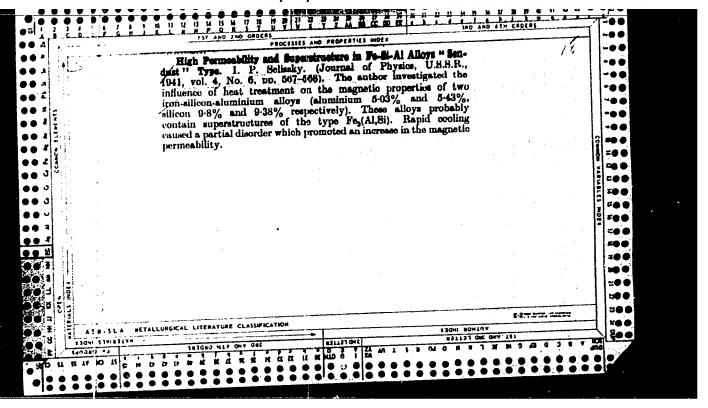






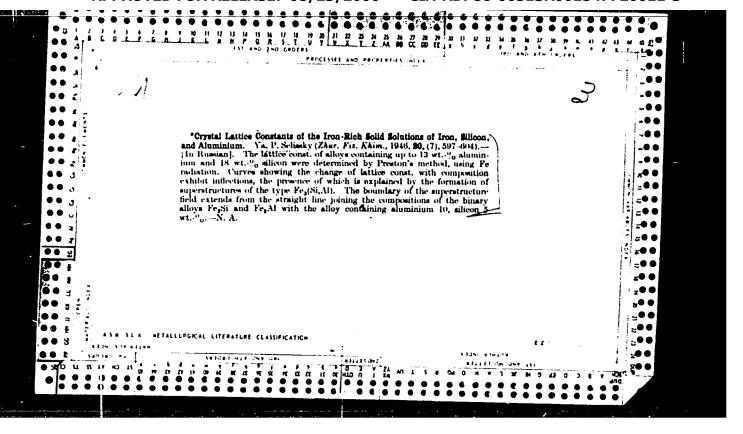


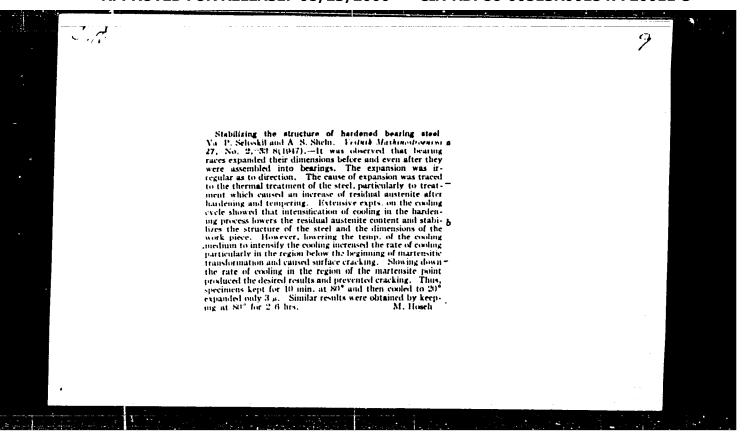


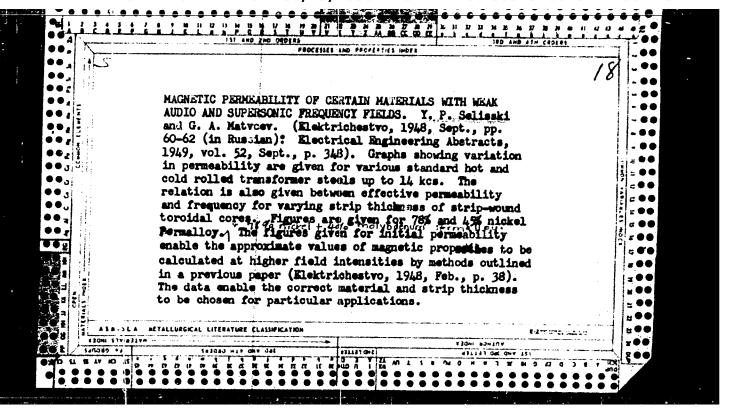


"Magnetic Materials in the Technics of Communications," Nauchno-Tekhnicheskiy Eulleten', MIIISSV, 7, 1, 1946

SELISSKIY, Ya. P.







SELISSKIY, YA. P.

PA 20/49T23

Dec: 48

USSR/Electricity

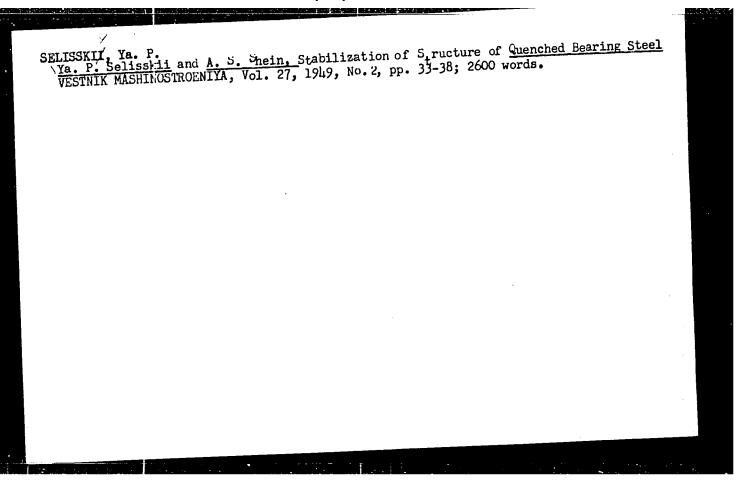
Transformers - Cores Coils, Choke

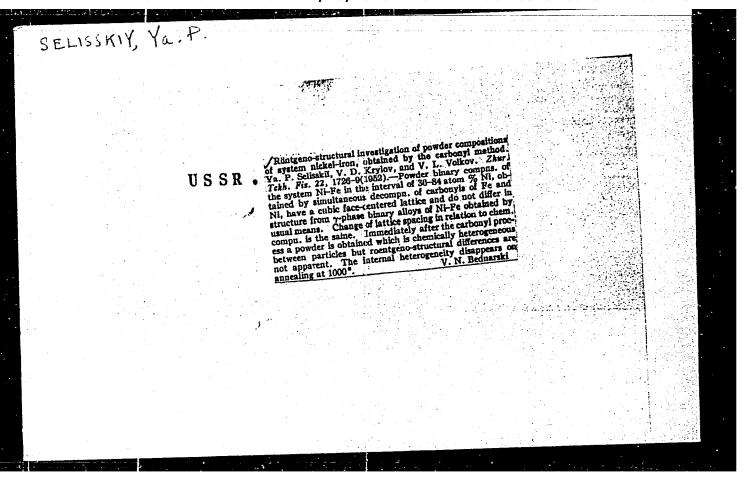
"Materials and Some Features of Belt Transformer Cores," Ya. P. Selisskiy, Cand Tech Sci, Sci Res Testing Inst for Communications of the Armed Forces imeni Voroshilov, 3 pp

"Elektrichestvo" No 12

Recent increases in demands for high-frequency transformers and choke coils have led to use of new materials for manufacture of cores and use of belt cores. Shows properties of materials and features of these cores.

20/49123





RUMYANTSEV, S.V.; GRIGOROVICH, Yu.A.; SELISSKIY, Ya.P., redakter; VAYNSHTEYN, Ya.B., tekhnicheskiy redaktor.

[Quality control of metals through the use of gamma rays]
Kontrol' kachestva metallov gamma-luchami. Moskva, Ges.
nauchno-tekhn. izd-vo literatury po chernoi i tsvetnoi metallurgii, 1954. 248 p.
(MLRA 7:8)
(Metals--Analysis) (Gamma rays)

HUME-ROTHERY, William; LYIBOV, B.Ya., redaktor [translator]; SELISSKIY,
Ya.P., redaktor [translator].

[Atomic theory for students of metallurgy. Translated from the
English Atomnaia teoriia dlia metallurgov. Perevod s angliiskogo

(Atomic theory) (Electrons) (Metals)

HUME-ROTHERY, W.; CHRISTIAN, I.W.; PEARSON, W.B.; KADYKOVA, G.N. [translator];
KRASNOPEVTSEVA, T.V. [translator]; RAVDEL', M.P. [translator];
SELISSVIV, Vo.P., redaktor; GOL'DENBERG, A.A., redaktor; ARKHANGEL'SKAYA, M.S., redaktor izdatel'stva; EVENSON, I.M., tekhnicheskiy
redaktor

[Metallurgical equilibrium diagrams. Translated from the English]
Diagrammy ravnovesiia metallicheskikh sistem. Perevod a angliiskogo
B.N.Kadykovoi idr. Pod red. IA.P.Selisskogo. Moskva, Gos. nauchnotekhn. izd-vo lit-ry po chernoi i tsvetnoi metallurgii, 1956. 399 p.

(Phase rule and equilibrium)

(Alloys) (Solutions, Solid)

SELISSKIY, YA.P.

G-4

USSR/Electricity - Conductors

: Referat Zhur - Fizika, No 5, 1957, 12239 Abs Jour

Author

: Gutovskiy, I.G., Selisskiy, Ya.P.

Inst Title : Anomaly of Electric Resistance in the Fe₃Si Alloy.

Orig Pub

: Fiz. metallov i metallovedeniye, 1956, 2, No 2, 375-376

Abstract

: Measurement was made of the electric resistivity (R) of an alloy Fe3Si at high temperatures (T) to determine the temperature region of the disordering of this alloy. As the critical temperature of ordering was approached, there should have appeared, in connection with the reduced degree of order, an additional resistance with a break on the R vs. T curve at the critical point. Measubreak on the R vs. T curve at the critical point. rements have shown a strictly linear course of the R vs. T curve up to 6000, where a sharp break in the curve was observed and a subsequent slight reduction in the electric resistivity as the temperature was increased to

Card 1/2

APPROVED FOR RELEASE: 08/23/2000 CIA-RDP86-00513R001547720011-3

: Ref Zhur - Fizika, No 5, 1957, 12239 Abs Jour

> 11000. The linear course of the R vs. T curve in the region from 20 to 600° indicates that in this region there is no disordering process, and this indeed explains the absence of an additional increase in R above that due to the scattering of conduction electrons by the thermal vibrations of the lattice.

KADYKOVA, G.N.; SELISSKIY, Ya.P.

Ordering process in iron-cobalt alloys. Fiz.met. i metalloved. 3
(MIRA 10:3)
no.3:486-496 '56.

1. Institut pretsizionnykh splavov TSentral'nogo nauchnorissledovatel'
skogo instituta chernoy metallurgii.
(Iron-cobalt alloys-Metallography)

KADYKOVA, G.N.; SELISKIY, Ya.P.

Ordering rate in FeCo alloys. Fiz.met. i metalloved. 3 no.3:497(NIRA 10:3)
502 '56.

1. Institut pretsizionnykh splavov TSentral'nogo nauchno-issledovateliskogo instituta chernoy metallurgii.

(Iron-cobalt alloys-Metallography)

ARTSISHEVSKIY M.A. [translator]; SELISSKIY, Ya.P., red.; GRYAZNOV, I.M., red.; ARKHANGEL'SKAYA, M.S., red.izdatel'stva; KARASEV, A.I., tekhn.red. red.; ARKHANGEL'SKAYA, M.S., red.izdatel'stva; KARASEV, A.I., tekhn.red. [Effect of nuclear irradiation on structure and properties of metals and alloys. Translations.] Deistvie iadernykh izluchenii na and alloys. Translations.] Deistvie iadernykh izluchenii na strukturu i svoistva metallov i splavov. Perevod M.A.Artsishevskogo, strukturu i svoistva metallov i splavov. Perevod M.A.Artsishevskogo, pod red.IA.P.Selisskogo. Moskva, Gos.nauchno-tekhn.izd-vo lit-ry pod red.IA.P.Selisskogo.

134

AUTHOR:

TITIE:

On the process of ordering in Fe Al alloys.

(O protsesse yaporyadocheniya v splace Fe Al.)

"Fizika Metallov i Metallovedenie" (Physics of Metals and Metallurgy), 1997, Vol. IV, To.1 (10), pp.191-192, (U.S.S.R.) PERTODICAL:

ABSTRACT:

The process of ordering in the alloy Fe Al was investigated by measuring the volume changes caused by the formation or the breaking up of a super-structure, and on the basis of the breaking up of a super-structure affects were investigated thermo-magnetic curves. The volume effects were investigated thermotrically and by X-rays (measurement of the period of the lattice of powders hardened from trailous temporatures the lattice of powders hardened from various temperatures the lattice of powders nardened from various bemperatures. On after step-wise annealing by means of a Preston chamber). On heating specimens hardened from 700 and 850 °C an appreciable reduction in volume occurs due to ordering from 180 °C. reduction in volume occurs due to ordering from 180 °C and onwards; it is most pronounced between 260 and 285 °C and observed in the second contract of the co changes sign at higher temperatures due to the beginning of disordering. The Curie point was measured at 610°C and at that temperature the dileteration current should be the beginning of the temperature the dileteration current should be the beginning of the temperature the dileteration current should be the beginning of the temperature the dileteration current should be the beginning of the be that temperature the dilatometric curves show a bend. 1 graph, 1 Russian and 3 English references.

Jerrous Metallurgy Research Institute. Recd.Oct.1, 1956.

581155A-14.W.P RAVDEL', M.P.; SELISSKIY, Ya.P.

Transformations in ternary Ni₃ Fe-base solid solutions. Dokl. AN SSSR 115 no.2:319-321 Jl 157. (MIRA 10:12)

1.Institut pretsizionnykh splavov TSentral'nogo nauchno-issledovatel'skogo instituta chernoy metallurgii. Predstavleno Akademikom I.P. Ba:dinym. (Iron-nickel alloys)

SELISSKI YA R.
AUTHOR Tran

TITLE

RAVDEL/ M.P. and Selisakiy, Ya.P.

Transformations in Ternary solid Solutions with Nize as Priner.

(Prevrashcheniya v troynykh tverdykh rastvorakh na Osnove Nize. Russian)

Osnove Nize. Russian)

Doklady Akademii Nauk SSSR 1957, Vol 115 Nr 5,

Doklady Akademii Nauk SSSR 1957, Vol 115 Nr 5,

PERIODICAL

ABSTRACT

In the first four papers quoted the special influence by molybdenum on the arranging alloys of a composition close pp 319-321 (U.S.S.R.) to Nizre was emphasized. It was found that alloying with molybdenum fundamentally changes the transformation character in the annealing of these alloys. This shows in an anomaly of electric resistance and in effects of volume deviating from the arrangement. In the present paper the influence of various elements & Mn, Si, Cu, No, V. and W - on the process of arrangement of the Ni Fe alloy was investigated. Ill. 1 shows the change of the Velectroresistance of alloys with various alloying additions as dependent on the quenching temper rature in the course of gradual cooling. The initial state of all alloys was obtained after quenching from 900°C in water. The duration was varied from 24 to 120 hours according to the quenching temperature.

CARD 1/4

Transformations in Ternary Solid Solutions with NigFe as Primer.

Alloying with 3 % Mn considerably intensifies the effect of arrangement in non-alloyed Nizke and markedly raises the temperature of the transfermation "order-disorder", Alloying with 5% Cu reduces this effect. It acts in the same way as the deviation from the stoichiometric composition in a binary alloy Fe-Mi (e.g. 79 % Mi). In the alloys with 45 % Cr, 4 % v and 4 % W an abnormal inorease in electric resistance develops after a longlasting gradual heat-treatment (6,5 % in Gr, 12 % in do, 20 % in Va). In the W-alloy it is comparatively small (2,5 %). In the Si-alloy it is 3 %. This alloy also shows the same anomaly. The curve here has a maximum at 450 %. Above that the resistance in S1 increases more intensively than in other elements. Ill. 2 records the surves of change of the coefficient of thermal expansions a as dependent on temperature. The initial state was reached in like manner as above, but between 250 and 550°C. Ill. 3 gives the thermomagnetic curves characteristic for alloys with only one element which latter exhibits an anomaly of electrorestistance. Ill. 4 Sives a compania son of the same curves of the arranging alloys Kizke and Ni3 (Fe, Mn).

CARD 2/4

20-2-34/52

Transformations in Ternary Solid Solutions with Ni3Fe

The thermomagnetic curves of the alloys which show the as Priner. anomaly of resistande and the abnormal course of the coefficient of thermal expansion, are similar and characterized by an indistrinct magnetic transformation on heating and coolong. The latter may be explained by the local chemical heterogenousness of the solid solution. Chemical complexes distinguished by a higher Curie point apparently form around the atoms of admixture at a certain temperature due to their chemical relation. ship with the atoms of the chief components. The more the alloying element differs from the chief elements the stronger is the chemical relationship and the stabler the developing complexes. The peculiar influence of Mn is apparently connected with the fact that, in a solid solution which possesses an incompletely built 3d-shell, Mn participates in the magnetic interaction. In an orderly arrangement the magnetic saturation of the N₃(Fe, Mn)-alloy is much higher than in Ni₃Fe, whereas

CARD 3/4

24-2-34/62

Transformations in Ternary Solid Solutions with Migre as Primer.

We after hardening reduces the magnetic saturation of Migre just as the other elements. An additional wedgemion of the electric resistance in annealing is therefore connected with the increase of magnetic saturation in the arrangement of the Mig (Na, Mn)-alloy,

(4 Illustrations, 2 Slavie referrences)

ASSOCIATION:

Institute of Precision Al ora of the Central Scientific

Research Institute of Farmers Metallargs.

(Institut pretsizionnyka spierov rasutrantago nauchno-

issledoratel skogo instituta chernov metal urgii)

PRESENTED BY:

Bardin, I.P., Academician, March 13, 1957

SUBMITTED: AVAILABLE:

12.3. 57 Library of Congress.

CARD 4/4

SELISSKIY

reports of an Inter-vuz Conference on Relaxation Phenomena in Pure Metals and Alloys

SOV-3-58-9-25/36

2-4 Apr 1958, Moscow Inst. of Steels. resilient reaction of spring alloys, Institute) covered the resilient reaction of spring alloys, various physical and technological effects on it and the methods of its measurement. Ya.P. Selisskiy (Institute of Precision Alloys TsNIIChM) told of subsiding oscillations of ultrasonic frequency in some ferromagnetic solid solutions. R.I. Garber and A.I. Kovalev (Physico-Technical Institute UkrSSR AS in Khar'kov) spoke of the temperature dependency of moduli of elasticity of iron.

Vest. Vysshe Shkoly, 0, 72-3, 1950 (Piguzov, Yu. V.)

Card 4/4

PHASE I BOOK EXPLOITATION SOV/3528 Outerdantonestry propagandy Outerdantonestry propagandy Intervention of the proposition o	
ANTION SOV/3528 be bornik statoy (In- Articles) Hoscow, Articles) Hoscow, Entitles and Mathematical Or Electure and In Nav. Poscovokia, Enginery Listenses, Radiators Nav. Lagania, Candidate Ergelliakia, Candidate Problema in Candidate of Candidate of Ergelliakia, Candidate Problema in Nemanicokia, Applicat Candidate Nav. Electric Cantenents Engersia Nav. Electric Nav. Radiators Nav. Englances, Effect of of Crystallization and Processi Irystallization and Processi Irystalli	175 184 211 223 240 'or 253
PRINCE I DON EXPLOINTION NAME AND ADDRESS OF PROPAGATION THE STATE OF THE STATE OF A S	Bagdasarov, Kn.S., Candidate of Chemical State of Intersonte Intersonte Ultrasonic Variations on the Process of Crystallisation Intersonte Ultrasonic Variations on the Process of Crystallisation Prakipatory Prakipatory Control of The Control of Prakipatory Versolove, M.N., Candidate of Technical Sciences. Ultrasonic Dor Trendidate of Technical Sciences. Ultrasonic Dor Technical Control of Case Depth in Electrically Control of Plakipatory Woodus Inspection of Case Depth in Electrically Margaray Steel Products Rabkin, N.V. Engineer. Design of Piczoelectric Transducers for Ultrasonic Plakipatory Plakipatory

and 126-7-1-7/28 sy, 6.2. koahelysyst, d.V. frame on kleetrical Beatstance sy yeld and the Ageing alloy trovki deyronami na elektro- rocki deyrocki na elektro- rocki na	Granteness on electrical resistance of the control	larger than the experients error, but that take sapisation positive in quenched samples and negative in aged samples. Further ordering of the Naye alloy. In the Feath alloy further ordering of the Naye alloy. In the Feath alloy during the disordered samples as are intermediate between the disordered and ordered samples as are intermediate between the disordered and ordering a samples are are intermediated to tempering at various temperatures. In the ease of Feath the amps of the aloring resistance is aspected at 2500, confirmed that destreen irreducting when the irreduces an intermediate of ordering. When the irreduces an intermediate of ordering. When the irreduces an intermediate of ordering was longitioned to reach a state of equilibrium of tempering were longitioned the non-irreduced persent he (Fig.3). We samples were tampered the of ordering was been intermediated and the non-irreduced persent he (Fig.3). We reposite the irreduced and the non-irreduced persent he samples after tempering. There are 2 figures, 3 tables and samples after tempering intermediated will be intermediated will (Institute of Institute of Institute of Alloys Tables in the control of the institute of Institute, Mondow State University).	
AUTHORS: Artsiabavally, M.A., Vasilyav, 6.3., Koshelyayav, 0.7. AUTHORS: And Selisatly, M.A., Vasilyav, 6.3., Koshelyayav, 0.7. Extract of Deuteron-Bonbardsant on Meetical Resistant of the Ordering Alloys Highs, Pe.All and the Ageing Alloys Highs, Pe.All and the Ageing Alloys Highs, Pe.Alloys of Deuteronasi na elektronogramming apparently bonbardireytahahithaya splavow Milghe, pe.Alloyin Pe.	ABSTRACT: The suture restriction of the Aging alloy with Toke Sizes which and state a few of 20-20, thickness which it is a state the series of 20-20, thickness which it is a state of 20-20, thickness which it is a state of 20-20, thickness which it is a translated by a supplied to writing forms of the translation by a supplied of the size of 2500/hour from 550-250 ordered state of the size of the size of 2500/hour from 550-250 ordered states of the size, we said alloys were profited by the size of the size of 2500/hour from 550-250 ordered states of the size of 2500/hour from 550-250 ordered states of the size	larger than the apprimental error, but their seed samples, positive in quenched samples and negative in aged samples. The positive in quenched samples and negative in aged samples for a the seed of	

SOV/126-7-2-9/39

Borodkina, M. M., Detlaf, Ye. I. and Selisskiy, Ya.P. . 24(2), 18(3), 18(7)

Recovery and Recrystallisation in the Ordering Alloys AUTHORS:

Fe-Co (Vozvrat i rekristallizatsiya v uporyadochiva-TITIE:

yushchikhsya splavakh Fe-Co)

PERIODICAL: Fizika Metallov i Metallovedeniye, 1959, Vol 7, Nr 2,

pp 214-224 + 1 plate (USSR)

The results of an investigation carried out with the aim of elucidating the characteristics of recovery of the initial stage of recrystallisation of Fe-Co ABSTRACT:

alloys in relation to cobalt content are described in this paper. Alloys, the compositions of which are shown in Table 1, were cast from Armco iron and

cobalt K-1 into ingots weighing 1 kg. These were forged at 1180°C into billets and subsequently rolled at 1100 to 1150°C into strip of 3 mm thickness. The

hot rolled strip was cut into squares which were water quenched from 900°C and cold rolled to thicknesses of 0.5 and 0.1 mm. Square specimens 20 \times 20 mm were

cut from the cold rolled strip. These were sealed in evacuated quartz ampules and annealed at temperatures

of: 150, 300, 400, 450, 500, 550, 600, 700 and 750°C, Card 1/6

SOV/126-7-2-9/39

Recovery and Recrystallisation in the Ordering Alloys Fe-Co

at which they were soaked for 5, 10 and 15 mins, 1 and 2 hours. In special cases the soaking time was'8 hours. Cooling was carried out in air. Specimens of 0.5 mm thickness were used for hardness tests on a Vickers machine using a load of 5 kg and for an X-ray investigation in a RKE camera for rapid exposure (Ref 2) and in a KROS camera. Exposure in this case was carried out in a Co irradiation, both the adaptor and the The distances between the specimen were rotated. object and the film was 100 mm. In the X-ray photographs the K_{α} -doublet lines from the plane (013) were visible which in the case of deformed specimens appeared diffuse and merged into the background of the X-ray picture. An increase in Co content of Fe-Co alloys above 25% is associated with a decrease in lattice parameter (Ref 3), as a result of which the doublet of (013) shifts in the direction of large Bragg angles of for a 25% Co alloy 3 = 81, for a 75% Co alloy 3 = 86. For this reason the sensitivity of the method to change in line width was great and increased with increasing Co content. In order to estimate the

Card 2/6

SOV/126-7-2-9/39

Recovery and Recrystallisation in the Ordering Alloys Fe-Co changes in width and intensity of the doublet line

during annealing, the X-ray films were photometered recrystallisation was indicated by the appearance of separate interference spots in the doublet line on exposure to the KROS camera with a rigid specimen and adaptor. Besides, specimens of 0.1 mm thickness were investigated in a Mo irradiation in a camera with a flat adaptor in order to obtain textural X-ray pictures at an object-film distance of 60 mm. Here the interference rings of the (011), (002) and (112) planes were clearly apparent, from which the nature of the texture obtained could be established and the progress of recrystallisation could be seen. In Fig 1 the annealing temperature and minimum soaking time required for the appearance of the maxima K and Ka in the photometric curve is shown in relation to the Co content of the alloy. Fig 2 shows microphotometric curves for alloys with different Co content which have been annealed at 400°C for 30 mins. Figs 3 Card 3/6 and 4 show micro-photometric curves for 65% Co and

sov/126-7-2-9/39

Recovery and Recrystallisation in the Ordering Alloys Fe-Go 35% Co alloys respectively which had been annealed at various temperatures and for various soaking times. Fig 5 shows micro-photometric curves for a 42% Co alloy which had been annealed at various temperatures for 2 hours. Fig 6 shows the temperature ranges of recovery and recrystallisation of alloys with differing Co contents: I - $K_{\alpha 1}$ and $K_{\alpha 2}$ maxima; II - sharp $K_{\alpha 1}$ and maxima; III-appearance of separate interference spots in the ring; IV - complete disappearance of the continuity of the ring. The region of supplementary diffuseness of the interference lines is indicated by brackets. On the basis of their experiments, the authors arrived at the following conclusions: 1) A relationship between the temperature range of recovery and the composition of the Fe-Co alloys investigated has been established. The beginning of

the breaking up of the K_{α} doublet in X-ray photographs, characterising the initial stage of recovery, is observed at very low temperatures in alloys of the stoichiometric compositions Fe₃Co, FeCo and FeCo₃. This

SOV/126-7-2-9/39

Recovery and Recrystallisation in the Ordering Alloys Fe-Co is due to the fact that in a number of solid solutions,

the ordering alloys after deformation are thermo-dynamically least stable.

2) All cold deformed Fe-Co alloys containing between 25 and 75% Co can harden on low temperature annealing. The hardening takes place at annealing temperatures which are not high enough to give a broken up doublet. This hardness is due to ordering in the non-uniformly stressed lattice and formation of mixed regions of a different degree of ordering. In spite of some increase in stress in the distortion of the lattice at various intervals of the ordering process which brings about

hardening, the process on the whole must lead to a decrease in free energy.

3) In alloys containing 35 and 42% Co the repeated diffuseress of the doublet coincides in towards. diffuseness of the doublet coincides in temperature with a retardation in the fall of hardness after attaining a maximum in hardness-annealing temperature curves (35% Co) or even with the appearance of a second maximum (42% Co). The effect described takes place in the transformation range which was found by Masumoto,

Card 5/6

SOV/126-7-2-9/39

Recovery and Recrystallisation in the Ordering Alloys Fe-Co

Saito and Shinozaki (Ref 4) by means of thermal capacity measurements.

4) Recrystallisation in the ordering Fe-Co alloys commences at order-disorder transformation temperatures, Recrystallisation commences at the highest temperature in an alloy of the stoichiometric composition FeCo. There are 9 figures, 2 tables and 6 references, 2 of which are Soviet, 4 English.

ASSOCIATION: Institut pretsizionnykh splavov TsNIIChM (Institute of Precision Alloys TsNIIChM)

SUBMITTED: May 14, 1957

Card 6/6

Selisskiy, Ya.P. AUTHOR:

On the Evidence of the Transformation in the FezAl

Alloy being a Second-Order Phase Change TITLE:

PERIODICAL: Fizika metallov i metallovedeniye, 1959, Vol 7, Nr 4,

pp 534-543 (USSR)

It has been proved by Rhines et al (Ref 1 and 2) that the order disorder transformations in alloys whose ABSTRACT:

composition is given by the formulae Cu3Au and CuAu are, in fact, classical phase transformations, i.e. firstorder phase changes. These findings, however, do not necessarily apply to all alloys in which the order disorder transformation occur and the object of the investigation described in the present paper was to

determine the character of the transformation taking place in the FegAl alloy. X-ray and dilatometric measurements were used for this purpose, the experimental

alloy containing (in weight %) 13.2 Al, 0.07 Mn, 0.11 Si, 0.025 C, the remainder Fe. The dilatometer specimens (50 mm long, 3 mm diameter) were machined from

8 mm diameter rods obtained by hot forging the cast ingots and annealed (in hydrogen) at 1100°C for 3 hours.

Card 1/11

On the Evidence of the Transformation in the Fe₃Al Alloy being a Second-Order Phase Change

The same rods were used for the preparation of filings used for the X-ray analysis. (The particle size of the filings used in the experiments did not exceed 5.10^{-3} mm.) Heat treatment of the X-ray specimens was carried out in a vacuum quenching furnace shown schematically in Fig 1. The filings were contained in a small quartz boat (1) which was suspended inside a quartz tube (2) on nichrome wires (3) attached to a permalloy, arc-shaped anchor (4). On the completion of the heat treatment, the permalloy anchor (4) was lifted from its holder (5) with the aid of a magnet, and the quartz boat with the filings was dropped into the quenching tank (6). Oil D-2 (used generally in diffusion pumps and characterized by low vapour pressure) was used as the quenching medium. No sintering of the iron-aluminium filings heat treated in vacuum of 5.10 mm Hg occurred at the highest temperature employed (850°C). On falling into the quenching oil, the filings formed a suspension; this ensured that all particles were cooled at sufficiently fast and uniform rates. In the first series of experiments all X-ray specimens were heated to

Card 2/11

On the Evidence of the Transformation in the FezAl Alloy being a Second-Order Phase Change

700°C, held at this temperature for 1 hour and then heated or cooled to the quenching temperature. The heat treating cycles are listed in column 1 of Table 1; the numbers following the temperatures denote the holding time (in hours, except the first cycle where the holding time at 800°C was 20 min) at the temperature; temperature in each cycle is that from which the X-ray specimen was quenched. The second column gives the values of the lattice parameter, a, of the corresponding specimens. The symbols $I_{(111)\alpha}$ and $I_{(220)\beta}$ in the expression the values of which are listed in column 3, denote the intensities of lines (111) α of the superstructure of the Fe3Al alloy and lines (220) $_{\beta}$ of the normal crystal lattice. The values of S listed in the last column of Table 1 were calculated from the data listed in column 3 using the formula at the bottom of p 537; these values give the relative measure of the degree of the long-range order at various temperatures. The temperature dependence of the lattice parameter of Fe3Al (plotted from the data in Table 1) is snown

Card 3/11

On the Evidence of the Transformation in the FezAl Alloy being a Second-Order Phase Change

graphically in Fig 2a. The rate at which the lattice parameter attains the equilibrium value is characterized by data given in Table 2. The heat treating cycles are given in column 1: all specimens were heated to 750°C and held at this temperature for 2 hours; the first 4 specimens were then cooled in 10 min to 410°C and quenched immediately or after 1, 4 and 7.5 hours' holding at this temperature; the last 4 specimens were cooled in 15 min to 350°C and then quenched immediately or after 1, 2.5 and 3 hours at this temperature. In the next stage of the investigation the order-disorder transformation was studied by means of dilatometric measurements, in which two series of specimens were used. All the specimens in the first series were subjected to the same preliminary heat treatment which consisted of 2 hours at 750°C followed by 30 min at 800°C and quenching in water. The dilatometric measurements were taken during both the heating and cooling cycles, the rate of heating being 300°C per hour; for the cooling cycle the dilatometer heater was switched off. The typical

Card 4/11

On the Evidence of the Transformation in the Fe3Al Alloy being a Second-Order Phase Change

results of these measurements are reproduced in Fig 2b in the form of differential dilatometer curves showing the difference (in mm) between the expansion or contraction of the investigated specimen and that of a standard specimen in which no solid state transformation took place. From the data reproduced in Fig 2b the temperature dependence of the coefficients of thermal expansion was plotted (10^{-6} /°C versus °C) for both the investigated and standard specimen (Fig 2v, curves 1 and 2 respectively). The dilatometer specimens of the second series were all quenched from different temperatures, having been first heated to 700°C and held at this temperature for 2 hours; while being cooled to its quenching temperature, each specimen was subjected to one or more (depending on the quenching temperature) isothermal treatments (each of 2 hours duration) at the following temperatures: 550, 500, 450, 350, 300 and 250°C. During the dilatometric measurements these specimens were heated at a rate of 200°C/h; on cooling, the specimens were held for one hour at each of the following

Card 5/11

On the Evidence of the Transformation in the Fe AlzAlloy being a Second-Order Phase Change

temperatures: 700, 550, 500, 450, 400, 350, 300 and 250°C; the rate of cooling between each of these temperatures was not faster than 5°C/min. The dilatometer curves of the specimens of the second series, quenched at 700, 550, 500, 450, 400, 350, 300 and 250°C are shown in Fig 3a, b, v, g, d, e, zh and z respectively. It was observed in the course of these experiments that on cooling, when the specimens were held at constant temperatures, the dilatometer reading was also constant which indicated that the volume changes occurring in the specimens due to the disorderorder transformation took place rapidly and that the state of equilibrium was already attained at the beginning of each isothermal treatment. The dilatometer curves of the specimen quenched from 700°C (Fig 3a) are similar to those shown in Fig 2b. Regarding the curves of other specimens, quenched from progressively lower temperatures, it will be seen that the lower the quenching temperature, the less pronounced is the minimum on the heating part of the dilatometer curve;

Card 6/11

On the Evidence of the Transformation in the Fe3Al Alloy being a Second-Order Phase Change

this is associated with the fact that the lower the quenching temperature, the higher was the degree of the long range order in a given specimen at the beginning of the experiment. This effect is shown also by the variation of the coefficients of thermal expansion calculated from the heating parts of the dilatometer curves in Fig 3, for three temperatures - 250, 275 and 300°C; the relationship between the expansion coefficients and the quenching temperature is shown in Fig 4. It will be seen that while the expansion coefficient of specimens quenched from 700°C is quite small (becoming negative at 275°C as a result of the large volume change due to ordering), it becomes larger as the quenching temperature decreases; specimens quenched from 250 to 400°C have the expansion coefficients practically the same at the three selected temperatures. The various temperatures indicated in Fig 2 and 3 are given the following interpretation: $T_1 = 180$ °C is the beginning of the non-linear expansion of the (quenched i.e. disordered) alloy on heating and of the decrease of

Card 7/11

CIA-RDP86-00513R001547720011-3"

APPROVED FOR RELEASE: 08/23/2000

On the Evidence of the Transformation in the Fe₅Al Alloy being a Second-Order Phase Change

the expansion coefficient; $T_2 = 260$ °C marks the beginning of rapid contraction of the alloy on heating and the change of the sign of the expansion coefficient from positive to negative; $T_{\rm pH} = 270^{\circ} {\rm C}$ is the temperature at which the order-disorder transformation begins in the heated alloy; T₃ = 280°C marks the end of rapid contraction of the heated alloy and the change of the sign of the expansion coefficient from negative to positive; T4 = 370°C is the temperature at which both the standard and the investigated specimens have the same thermal expansion coefficients; TKN1 545°C marks a deflection on the heating portion of the dilometer curve, a sharp maximum on the graph of the temperature dependence of the expansion coefficient and disappearance of the long range order; $T_c = 610$ to 615° C marks the magnetic transformation; T_{KOl} deflection point on the cooling portion of the dilatometer curve whose position depends on the rate of cooling and which indicates the appearance of the long range order; $T_{yo} = 300^{\circ}\text{C}$ marks the beginning of the linear contraction

Card 8/11

SOV/126-7-4-7/26

On the Evidence of the Transformation in the FezAl Alloy being a Second-Order Phase Change

Thus, when the Fe3Al alloy of the alloy during cooling. in which the disordered structure has been retained by quenching is heated, the disorder-order transformation begins at a comparatively low temperature $(T_1 = 180^{\circ}C)$ and takes place within a comparatively narrow temperature range (180-270°C) while the order-disorder change occurs (on heating) within a wider temperature range (270-545°C). The disorder-order transformation in a specimen cooled from high temperature occurs between 545 and 300°C. The values of the lattice parameter (a) measured in the 550-250°C temperature interval (see Table 1) were used for the determination of the value of relative compression, $\delta a/\Delta a$, brought about by the (∆a is the total disorder-order transformation. reduction of the lattice parameter of alloy cooled slowly between 550 and 250°C; ba is the reduction of the lattice parameter cooled slowly from 550°C to a given quenching temperature) The relationship between 6a/\(\Delta\)a and $T/T_{\rm k}$ (where T is the absolute quenching temperature and $T_{\rm k}$ corresponds to absolute $T_{\rm KN1}$) is shown in Fig 5. both for the FegAl alloy (curve 1) and for the CugAu

Card 9/11

CIA-RDP86-00513R001547720011-3"

APPROVED FOR RELEASE: 08/23/2000

On the Evidence of the Transformation in the Fe₃Al Alloy being a Second-Order Fhase Change

alloy (curve 3). Curve 2 in Fig 5 represents the variation of S (the magnitude of which is proportional to the degree of the long range order) with $T/T_{\mathbf{k}}$. It will be seen that while there is a sudden change of volume of the Cu₃Au alloy at the Kurnakov point $(\tilde{\mathbb{T}}_{2a\perp})$, this being one of the characteristics of a first-order phase change, the volume of the Fe3Al alloy changes monotonically. The divergence between the S and ba/ a curves for the FegAl alloy is attributed to the fact that, according to Owen and MacArthur (Ref 6), the volume changes are associated with the initial stages of the disorder-order transformation and take place in a short time interval, while the variation of the intensity of the superlattice lines is associated with the growth of the anti-phase domains which is a slower process. It is stated in the concluding remarks that the absence of any discontinuities in the variation of the studied properties of the FezAl alloy during the disorder corder transformation, taken in conjunction with other published data, is a convincing proof that this transformation is a

Card 10/11

SOV/126-7-4-7/26

On the Evidence of the Transformation in the Fe3Al Alloy being a Second-Order Phase Change

second-order phase change. There are 5 figures, 2 tables and 10 references, 9 of which are English and 1 Soviet.

ASSOCIATION: Institut pretsizionnykh splavov Tsentral'nogo nauchnoissledovatel'skogo instituta chernoy metallurgii

(The Precision Alloys Institute of the Central Ferrous

Metallurgy Research Institute)

SUBMITTED: June 11, 1957

Card 11/11

AUTHCRS: Ravdel', M.P. and Selisskiy, Ya. P. SOV/126-7-6-13/24

TITLE: Investigation of Transformations in Alloyed Permalloy

PERIODICAL: Fizika metallov i metallovedeniye, 1959, Vol 7, Nr 6, pp 885-892 (USSR)

ABSTRACT: The authors have carried out a systematic investigation of hardening of Ni₃Fe-base alloys. The alloying elements used were Mo, Cr, Cu, V, W, Si and Mn and were added to the NizFe alloy at the expense of iron. The chemical composition of the alloys investigated is shown in a table, p 886. The alloys were melted in a high-frequency induction firmace and cast into ingots of 5 kg, homogenized in hydrogen at 1100°C and subsequently forged partly into billets and partly into rods of 8-9 mm. Specimens for dilatometric and thermomagnetic study (ℓ = 50 mm, d = 3 mm and ℓ = 25 mm, d = 3 mm, respectively) and wire of 1 mm diameter were produced from the rods. All electrical resistance measurements were carried out at room temperature on specimens of 1 mm diameter by a potentiometric method. The dilatometric study was carried out on a differential dilatometer of the Shevenar type which was Card 1/5 provided with a special device to enable isothermal soaking

sov/126-7-6-13/24

Investigation of Transformations in Alloyed Permalloy

to be carried out. The thermomagnetic study was carried out on an Akulov system anizometer. All specimens of the alloys investigated were subjected to stepwise heat treatment (heating to 900°C followed by stepwise cooling, with lengthy soaking at the following temperatures: 550, 500, 450, 400, 350, 300 and 250°C). After soaking, the duration of which varied between 24 and 120 hours depending on temperature, the specimens were immediately quenched in water. Such heat treatment ensured different degrees of order in the specimens in relation to the temperature of quenching. In Fig 1 the change in electrical resistance of a specimen quenched from 900°C in relation to the quenching temperature is shown. In Fig 2 dilatometric heating and cooling curves of a non-alloyed NizFe alloy, converted to the ordered state by stepwise heat treatment, are shown. In Fig 3 curves are plotted for the dependence of thermal expansion on temperature. Fig 4 shows the change in volume of alloyed Fe-Ni alloys during isothermal tempering in the dilatometer furnace. The tempering temperature was 450°C and the soaking Card 2/5 time 5 hours. Fig 5 shows thermomagnetic heating and

sov/126-7-6-13/24

Investigation of Transformations in Alloyed Permalloy

cooling curves. In Fig 6 similar curves are shown for the alloy Ni3 (Fe, V), containing 4% V. The authors arrive at the following conclusions: the introduction of 3% Mn at the expense of iron brings about a considerable intensification of ordering effects in a NizFe alloy. This is shown by a considerably increased drop in the electrical resistance after stepwise heat treatment. A greater volume effect can be observed in this alloy than in the selected one during disordering and isothermal The order-disorder transformation temperature of this alloy is higher than that of the NizFe alloy. The magnetic saturation of the ordered Ni₃(Fe,Mn) alloy is considerably greater than that of the NizFe alloy, whereas Mn lowers the magnetic saturation of a disordered NizFe alloy. Such peculiar influence of Mn is due to Other solid solutions, the structure of its 3d-shell. which also contain Mn, become ferromagnetic after It appears that Mn in the ordered lattice also participates in a magnetic reaction, as a result of which magnetic saturation increases strongly. A supplementary Card 3/5 fall in electrical resistance on tempering is associated

sov/126-7-6-13/24

Investigation of Transformations in Alloyed Permalloy

with an increase in saturation of the ordered alloy $Ni_{\chi}(Fe,Mn)$. Copper brings about a concentration disorder in the Ni₃Fe alloy. Additions of Cr, W and V act on the ordering process of the NizFe alloy in the same way as Mo (Ref 3), changing the nature of the effects. An anomaly in electrical resistance has been found to exist in Mo permalloy as well as in alloys containing Cr, V and W, i.e. an increase in the electrical resistance after heat treatment in the temperature range at which ordertakes place. All these alloys exhibit identical No lattice dilatometric and thermomagnetic anomalies. contraction, characteristic for the ordered state, occurs in these alloys. Also there is no sharp volume increase on disorder establishment. The dilatometric peculiarities of these alloys are characterized only by a change in the thermal expansion coefficient at the transformation temperature. In all these alloys a temperature range is observed for the ferromagnetic transformation instead of a sharply defined Curie point. Such an effect of Mo, Cr, W and V is due to the fact that complexes are formed in a Card 4/5 one-phase solid solution at definite temperatures, the

SOV/126-7-6-13/24

Investigation of Transformations in Alloyed Permalloy

Curie points of which differs from that of the basic solid solution. This can be seen even more convincingly in alloys containing. Si which forms very stable complexes. There are 6 figures, 1 table and 8 references, 1 of which is Soviet, 2 English, 3 German and 2 French.

ASSOCIATION: TsNIIChM

SUBMITTED: May 15, 1957 (Initially)

June 9, 1958 (After revision)

Card 5/5

3614HC s/137/62/000/003/105/191 A060/A101

12.8100 AUTHORS:

Artsishevskiy, M. A., Vasil'yev. S. S., Koshelyayev, G. V.,

Selisskiy, Ya, P.

TITLE:

Action of deuteron irradiation upon the electric resistance of

alloys undergoing ordering and aging

PERIODICAL:

Referativnyy zhurnal, Metallurgiya, no. 3, 1962, 6, abstract 3138

("Sb. tr. Tsentr. n.-i. in-t chernoy metallurgii", 1959, no. 22,

168-176)

The effect of deuteron irradiation upon the electric resistance R of alloys Ni₃Fe, Fe₃Al undergoing ordering and of an alloy of Fe with 35% Ni and 4.5% Ti undergoing aging was investigated. The specimens were irradiated in a cyclotron with deuterons having an energy of 4 Mev. The thickness of the TEXT: specimens constituted 20 - 30 10. The R measurement was carried out by the potentiometric method. Because of the small dimensions of the specimens the voltage and the current leads constituted a single whole with the working part. The specimens of NigFe and of FegAl were investigated in the ordered and the unordered states; the specimens of Fe-Ni-Ti - in the aged and hardened states.

Card 1/2

S/137/62/000/003/105/191 A060/A101

Action of deuteron irradiation ...

It was established that when the ordered $Fe_{\gamma}Al$ alloy is irradiated its R is increased considerably, and the R of the hardened alloy - drops. The bombarding of the Ni3Fe alloy in the ordered and unordered states causes a considerable decrease in R. In all cases irradiation in fluxes up to $5 \cdot 10^{17}$ deuterons per $1~\mathrm{cm}^2$ causes a sharp change in R, at a further increase of the total flux the rate of change of R drops. The effects uncovered in the Fe-Ni-Ti alloy do not exceed the limits of experimental errors. It is considered that the most probable process causing the reduction in R is the ordering. A considerable drop in the R of the alloy NigFe is noted, whose degree of ordering corresponds to a temperature of 250 - 300°C. In this alloy a further occurrence of ordering under irradiation is possible. The shape of the R curves of the irradiated specimens tempered at $250\,^{\circ}\text{C}$ confirms the hypothesis as to the attainment of an intermediate degree of ordering as result of the irradiation. In tempering the NigFe the soaking time of the specimens at the respective temperatures was insufficient to obtain an equilibrium. The character of the R variation of an irradiated unordered specimen is close to the R variation of an unirradiated ordered specimen. In tempering the Fe-Ni-Ti alloy no great difference in the behavior of irradiated and unirradiated specimens was discovered. A. Rusakov

[Abstracter's note: Complete translation]

Card 2/2

24 (2), 24 (6)

Borodkina, M. M., Detlaf, Ye. I.,

sov/48-23-5-22/31

AUTHORS:

Selisskiy, Ya. P.

TITLE:

X-ray Investigation of Interrelation in Processes of Recovery, of Recrystallization and of Ordering in the Alloys Fe-Co and Ni-Fe (Rentgenograficheskoye issledovaniye vzaimosvyazi protsessov vozvrata, rekristallizatsii i uporyadocheniya v splavakh Fe-Co i Ni-Fe)

PERIODICAL:

Izvestiya Akademii nauk SSSR. Seriya fizicheskaya, 1959, Vol 23, Nr 5, pp 640 - 642 (USSR)

ABSTRACT:

The increase of free energy in low-temperature deformation by tensions of the 2nd kind and the increase of the surface tension occur in consequence of texture destruction. For a number of solid solutions, the increase of free energy is related to the stoichiometric energy. These relations are shown in a diagram (Fig 1), in which the solid solution consists of the components A and B. In the case of low-temperature deformation, an increase by the quantity Δ E occurs in the free energy of the

solid solution which differs considerably from the stoichicmetric composition AB. The free energy of the stoichiometric

Card 1/3

X-ray Investigation of Interrelation in Processes of SOV/48-23-5-22/31 Recovery, of Recrystallization and of Ordering in the Alloys Fe-Co and Ni-Fe

composition changes by Δ E $_{n}$, and the total change of free energy is equal to the sum of both these quantities. Thermodynamic considerations are then made of the recovery, recrystallization and ordering. Next, the results of the radiographic investigation of the recovery and recrystallization of the alloys in question are dealt with. The relationship between recovery and the tensions of the 2nd kind and the distortions of the 3rd kind, revealed by an amplification of the radiographic lines, is made use of. A diagram (Fig 2) shows the microphotometrically plotted curves of the K_{α} doublet for three Fe-Co alloys, annealed for 30 minutes at 400°C. From the shape of these lines conclusions are drawn as to the stage of recovery. Figures 3 and 4 show series of roentgenograms of the alloys Fe-Co and Ni-Fe, annealed at various temperatures and different compositions. Conclusions as to the stage of recrystallization are drawn on the strength of the interference spots observable here.

Card 2/3

CIA-RDP86-00513R001547720011-3 "APPROVED FOR RELEASE: 08/23/2000

X-ray Investigation of Interrelation in Processes of SOV/48-23-5-22/31 Recovery, of Recrystallization and of Ordering in the Alloys Fe-Co and Ni-Fe

There are 5 figures and 5 references, 2 of which are Soviet.

ASSOCIATION: Institut metallurgii im. A. A. Baykova Akademii nauk SSSR

(Institute of Metallurgy imeni A.A. Baykov, Academy of Sciences,

USSR)

Card 3/3

8/137/61/000/010/030/056 A006/A101

Detlaf, Ye.I., Selisskiy, Ya.P. AUTHORS:

On the correlation of ordering, resovery and recrystallization pro-TITLE:

cesses in Fe-Co alloys

Referativnyy zhurnal. Metallurgiya, no. 10, 1961, 42, abstract 10Zh264 ("Sb. tr. Tsentr. n.-i. in-t chernoy metallurgii", 1960, PERIODICAL:

no. 23, 224 - 227)

An investigation was made with Fe-Co alloys containing up to 20-75% Co. The alloys were subjected to cold rolling with 83.5% total deformation by the 0.5 mm thickness and subsequent annealing at 150 and 750°C in a vacuum at different duration of heating. Subsequently the specimens were electropolished, Hy was measured, and X-ray examination was carried out by the method of reverse exposure on Ka _Co radiation. The lowest temperatures of recovery were observed in alloys of stoichiometric composition. If a decrease of free energy, connected with recrystallization, exceeds the increase of free energy connected with seftening, then recrystallization has a greater thermodynamical advantage and the

Card 1/2

"APPROVED FOR RELEASE: 08/23/2000 CIA-RD

CIA-RDP86-00513R001547720011-3

On the correlation of ordering, ...

S/137/61/000/010/030/056 A006/A101

recrystallization process may cause softening. In the opposite case, softening advances recrystallization. In alloys with 35, 42 and 50% Co both cases occur; this is manifested in the partial splitting-up of the doublet.

P. Zubarev

[Abstracter's note: Complete translation]

Card 2/2

"APPROVED FOR RELEASE: 08/23/2000 CIA-RDP86-00513R001547720011-3

Money. Permitting the minimization of the mini	!!
Printing any splay (Preises Ally) (Series: Its: Borrit rule, 779, 2) Fill. D., Galishus A. of Fallaning Borrita. M. D., Galishus A. of Fallaning Borrita. M. D., Galishus A. of Fallaning Borrita and the standard of sell- M. M. D., Galishus A. of Fallaning Borrita. M. M. D., M. M. D., M. O., M. O., M. O., M. O., M. M. M. D., M. M. D., M. M. M. M. M. M. M. M. M. M. M. D. M. M. M. M. M. M. M. M. M. M. M. M. M. M. M	y institut chemoy metallungit.
Maintent Spensories Agency USES, Constitution St. 19, Namely 20, 19, 19, 19, 19, 19, 19, 19, 19, 19, 19	now, Metallurgislat, 1960. 263 p. hrata alip inserved. 2,525 copies
Hair Dif Gentelyen Ed. of Philabale B Berlingth of States and the section of section of the section of section of section of the section of the section of the section of section section of section of the	Cosularatvennaya planovaya komissiya,
FERREZ: This book is intended for engines with a for inherital promote erganishment, it may also be useful promote organishment in the will as for inherital promote of the will solderston conducted in resent prace by the Central Promote Scientishment (Trainer) and the section of the Central Programmers and in Edd-Traylory suggestion of the Proposition of Proposition of the Proposition of Proposition of Proposition of Proposition of Proposition of the Proposition of Proposition of San Proposition of P	: Ye.I. Lavit; fech. Ed.:
corrected in recent pass by the Central Principe Control	and sitentific personn's in the factorish-equipment intustries, i in the production of presistin strending alvance! technical schools;
Description, Dit, and Got, Kaipen, Description, Description, 1714, and Got, Kaipen, Description, 1714, and A.S. Kaipen, Description, 1714, and M.S. Kaipen, Magnetic College, 1715, and M.S. Karlyon in the Magnetic College, 1715, and M.S. Karlyon in the Magnetic Remarkable on Their Macrocycle Description of Magnetic Library College, 1715, and M.S. Karlson, Description of Macrocy, 171, and M.S. Karlson, Macrocycle College, 1715, and M.S. Karlson, Macrocycle College, 1715, and M.S. Karlson, Macrocycle College, 1715, and M.S. Karlson, Macrocolle Macrocy, 171, and M.S. Karlson, Macrocolle Macroco	the fitting and the stigations of the stigations of the stigate of
Popular Core Popular V.J. Intestigation of the Propriet Soft Regalite Alloys of Various Indicates Soft Regalite Alloys of Various Indicates Mainty Dealer Conditions of Eigh-Lamitton V Mainty Mayorite Fields (13)-13) organist Mayorites I.M., and G.M. Experts. Besument Mayorites I.M., and M.M. Pradors, Iorginatin Mayorites I.M., and M.M. Pradors, Iorginatin Mayorites Alloys in the Ior-Crapterance Rase Mayorites (Iron-Mindlem Alloys Mayorites of Iron-Mindlem Alloys Mayorites of Mayorites of Protraction Frequest Mayorites of Mayorites of Son Heis-Taures I Maybern of Mayorites of Son Heis-Taures Mayorites of Mayorites Mayorites Mayorites Manana, Shot, Sh. Mayorites Manana, Shot, Sh. Mayorites Manana, Shot, Mananana Mayorites Mananan, Shot, Mayorites Mananan, Mananan, Mayorites Mananan, Shot, Mananan Mananan, Mananan Mananan, Mananan Mananan, Mananan Manan	
durith, Jail, Department of Drade Personalist Carrier, Yail, Dynade Magnetic Extragrinalist Charters of Edulation in Incomplete Allays in the Low-Experience Range and The Mary, I.W. Protects and Mary Mary, and Allays of Horal Law Mary, I.W. Protects and Horal Edulation of Protection Properties of Inco-Almadoms Allays from Experties of Inco-Almadoms Allays and Allays and Mary of Magnetics of Protection Theory of Magnetics of Protection of Son Health of Light-Present of Magnetics of Magnetics of Magnetics of Magnetics of Magnetics of Magnetics and Mary Domination of Magnetics of the Kinetic Edulation of Magnetics of the Kinetic Edulation of Magnetics of the Kinetic Edulation (Edulation of Magnetics of the Kinetic Edulation of Magnetics of the Kinetic Edulation (Edulation of Magnetics of the Kinetic Edulation (Edulation of Magnetics of the Kinetic Edulation (Edulation of Magnetics of the Kinetics).	55 and Strandary of 66
denvich, Ya.I. Dynamic Magneti Characteri Alloys Deder Conditions of Eigh-Lehaction Y madrova, L.G. Behavior of Certal Propers Eigh-Trejuery Magnetic Fields (195-10) cys Figher L.H., and G.J. Express. Setural anguestic Alloys to the Interpretation for Properties of Low-Trepretation in Prop. F.L., and J.M. Pederor. Lorgitudin in Prop. J.M. Investigations of the Europy of Misch. Mischell Alloys Richard Alloys Alloys and H.V. Moletillar, Magnetost Milpharry, L.M., and H.V. Moletillar, Magnetost Milpharry, L.M., and M.V. Magnetost Milpharry, L.M., and M.V. Schleider Froperties of Inter-Almafors Alloys Froperties of Inter-Almafors Alloys Denvities, M.M. Treura-Ashrit Attachmen for Errestigation of Deformation Treures i Bordetics, M.M., Treura-Ashrit Attachmen for Errestigation of Deformation Table; Bordetics, M.M., Treura-Ashrit Attachmen for Errestigation of Deformation and Eff. Emil- Eigh-Persobaltity Fron-Almafors Alloys Milpharms of Magness Manasco, Shir, Investigation of the Kinetic Fanasco, Shir, Investigation of the Kinetic	
Madrzory, I.O. Behavior of Gerald Ferrora High-Frichers (12.1.) cy Pedrzory, I.E., and G.A. Zyraevs. Bennests Magnetic Alloys in the Investments Rang Proves, V.P., and H.W. Pedrzor. Inregitation in Iron-Sitesh Alloys Nasy, I.W., and H.W. Molotilov. Magnetost Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama Alloys Malyzkama (1.7., and B.E. Belleidly Mary Malyzkama (2. Iron-Mimiliana Alloys Mary of Nasy cistication of Son Interve all Malyzkama (2. M. Freture-Atalysis Attachmen Malyzkama (2. M. M. M. M. Malyzkama Malyzkama Malyzkama (2. M.	
Pederor, L.B., and G.A. Zaytsers, Saumati magnit Alloys to the Investmenter Rang Indonesia in Iron-Sitab Alloys. Nasy, I.W., and B.Y. Neboor, Ingitadin in Sitab. Nasy, I.W., and B.Y. Neboor, Lagrany of Sitab. Nabolitor, B.Y., i.W. Parsy, and A.I. Bair, at Iron-Sitable Shykema Alloys. **Ribolitor, B.Y., i.W. Parsy, and A.I. Bair, at Iron-Sitable Shykema Alloys. **Properies of Iron-Mimdres Alloys. **Ribolitor, B.Y., Treurs-Kalyis Attachment for Iron-Sitable Shykema Alloys. **Ribolitor, B.W., Treurs-Kalyis Attachment for Iron-Sitable Shykema Alloys. **Ribolitor, B.W., Treurs-Kalyis Attachment for Iron-Sitable Shykema of Magnesia and Table Shopping of Magnesia and Shopping Shopping Shykema of Magneses. **Ribolitor, B.W., S. A. Makhama, Sail K.Y. Emil. **Ribolitor, B.W., S. A. Makhama, Sail K.Y. Emil. **Ribolitor, B.W., S. A. Makhama, Sail K.Y. Emil. **Ribolitor, B.W., S. A. Makhama, S. M. K. Emil. **Ribolitor, B.W., S. M. Makhama, S. M. M. Emil. **Ribolitor, B.W., S. M. M. M. E. M. M. M. E. M. Elmit. **Ribolitor, B.W., S. M.	tic Materials in Weak 108
Interpret this, well this responsibility in Tronsited Alloys Mary, I.M., and B.V. Molottlor, Magnetost Maryberry, I.M., and B.V. Molottlor, Magnetost Maryberry, I.M., and B.V. Molottlor, Magnetost Maryberry, I.M., and R.V. Soldstory Maryberry, I.M., Tretter-Assaysts Attachment for Investigation of Deformation Trettures 1 Morral of Deformation Trettures 1 Morral of Deformation of Sons Traffer Fright-Personal Reviews 1 Morral Maryberry of Magnetose Markberry of Magnetose Manage, Shift Investigation of the Kinetti Restification of Contraction Alloys Don Magnetose	Magnetization of Perro-
Michal Phasy 1. M., and B.V. Molotilov, Magnetoni Milydenina Alloys M. Party, and A.I. Halls of Iron-Bital-Molydenina Alloys Major Milydenina Alloys Milydenina Alloys Milydenina Alloys Milydenia of Iron-Almaina Alloys Magnetical of Iron-Almaina Alloys Magnetical of Deformation Teatures for Deformation Teatures Milydenia of Magnetical of Magnetical of Magnetical of Magnetical Milydenia of Magnetos Milydenia of Magnetos Milydenia alloys Don Malance Milydenia of Magnetos Milydenia of Magnetos Milydenia of Magnetos Milydenia Alloys Don Malance Milydenia of Magnetos Milydenia of Mi	Ų
*Molotilor, B.V., I.M. Pusey, and A.I. Rad's at Iron-Sinkel-Molytherma Alloys Physicals of Iron-Almansa Alloys Morphitas of Iron-Almansa Alloys Morphitas, M.M. Texture-Assigns Attachment for Errestigation of Priorestion Textures at formy of Magacian of Priorestion Textures at the Propy of Magacian coldination of Sum: Fight, Strong Light, Properties and Coldination Alloys Confinition of Magacian of Magacian Alloys Confinition of Magacian Alloys Confinition of Magacian Alloys Confinition of the Kinetic Ensembly Confinition Alloys Confinition of the Kinetic Ensembly Confinition Alloys Confinition of the Kinetic Ensembly Confinition Alloys Confinition (Confinition Alloys Confinition Alloys Confinition Alloys Confinition Alloys Confinition (Confinition Alloys Confinition Alloys Confinition Alloys Confinition (Confinition Alloys Confinition	139 tion of Mekel-Tron-
Allybers, I.F., scd Te.F., Schleidy, Magnities of Iron-Almedran Alloys Depolities, M.M. Treurs-Ashytis Atsolmen for Irrestigation of Protraction Treurs at Borditas, M.M., Z.M., Bullybers at I.f., 2. Borditas, M.M., Z.M., Bullybers at I.f., 2. Borditas, M.M., Z.M., Bullybers at I.f., 2. Inthits, B.G., M. Lakiman, sol K.F., Emiliar-Presidents of Naganese Estates of Naganese Estates Alloys Irrestigation of the Kinetic	
Dorolides, N.W. Texture-Arabysis Attachmen for irrestigation of Deforaction Textures i borolides, N.M., E.M. Buly-hers and Td.P., S ropy of Napelasticition of Show 1957-11279 Lightles, B.O., NO. Latinangasi, E.W. Emiliah- Malyderms of Napelasses Malyderms of Napelasses Equaso, Sh.L. Irrestigation of the Kinetic	Magnetostriction and Bone Other 166
borotitis, M.M., E.M. Bulyneva and felt. S royy of Nagacistativition of Some Institution Livatics, S.O., So. Laximan, sail E.V. Emil Experimentally Inco-Almarina alloys Con- Molyterms of Naganess Zanaso, Sh.I. Investigation of the Einstein	or the UES-SOL X-Ray Machine Line Alloy Thin Strip
Migh-brimeability iron-Almeira Alloys Cort Molyterim or Magnesse Zasana, Shii. Investigation of the Kinetti	sakir. Texture and Anisot- ing. Igrestigation of
TO THE PARTY OF TH	ning Additions of 194
Magnetic Texture in 05% Primalloy During In	respectation Anice alities 204
Endypers, Q.J., and [a.F., Sellsenly. Dilatometric Levestigation of Iron-Cobalt Alleys	ric Investigation of Iron-
Detlat, No. 11, and Na.P. Mallanir, Interrolation between the Ordering, Recovery, and Revretallisation Processes in YeCo Alloys	tion between the Ordering, 22% e-Co Alloys

"APPROVED FOR RELEASE: 08/23/2000 CIA-RDP86-00513R001547720011-3

SELISS KIÝ, YA.	Modeov. Institut stall **Redeov. Institut stall **Relationary professional professional professional professionary from Transportational professional professio	Sponsoring Acney: Ministeratro vysniego i srchiego syciolal'nogo obrazoranjyn. ESFSR and Moskoraldy institut stall incul IV. Stalina. Ed. (Trito pace): B.M. Finbal'shteyn; Ed, of Publishing Bouse: Te.1. Lorit; Tech. Ed.; A.I. Edrasor. Full A.I. Edrasor. Full of physical pacettion of articles is intended for pysnomel in selectific institutions and schools of higher education and for physical resolutions at a tritions and schools of higher education and for physical resolutions of these fields.	COVENUE: The collection contains results of experimental and theoretical iters- tigntions carried out by schools of higher clearing and strainfor rivers, institutions in the field of the relaxation phonouse in actals and allows, lastitutions in the field of the investigation—by the Internal-Critical Berearl articles are deroted to the investigation—by the Internal-Critical substitution that decrease of the experiments and lastic plants described in the actual infection at the investigation of substituting product, and the relaxion between internal fraction and tempor brittlesmest, the use of the scenario of their actual fraction and tempor brittlesmest, the use of the scenario of their actual fraction and tempor brittlesmest, the use of the scenario of their derivers of their actual articles of their actual fraction of materials, slautic affection and the now slow-detection method. No personalities are sentioned. References follow not articles. There are yet	Relystion Renomens in Metals (Cont.) Equal 2.12., and Den Ge-Sen [Inna'Korally gouderstrency university: [Kinn'kor Bitte University]]. Analysis of the Internal Friction of Powier- Mich. Earl [Institute of Technical Physics of the Cachoslorak Acaimy of Toriones). Mapertonecharical Phenomena in the Alternating Magnetic Field as a Relaxation Process [Splinkly, Ind. [Technical Typy nauthhoris*ledoratel'sky institut chernoy matchingil (Gentral Scientific Pescarch Institute of Perroms Metallury)). Magnetocarticiton, Hochins of Elastifict, and Internal Friction of Certain Iron-Base Perromagnets Soils Collity, and Internal Friction of Certain For-Base Perromagnets Soils Collity, and Metallury (Urrainal Scientific Research Intitute of Metalls)]. Study of the Impoct-Patigue Mehanism by the Dempting settled of Mesalls).	74/11/3126; 11817477 of Congress 74/11/3126; 71-25-61